



A systematic data collection on chloride-induced steel corrosion in concrete to improve service life modelling and towards understanding corrosion initiation



1. Introduction

Chloride-induced corrosion of reinforcing steel in concrete is one of the most frequent deterioration mechanisms for concrete structures [1], causing high costs to society [2,3]. Despite this, the factors controlling corrosion initiation of steel in concrete are still poorly understood [4]. The established concept to tackle this problem in science and engineering is based on a chloride threshold value for corrosion, the so-called critical chloride content (C_{crit}), where corrosion is assumed to occur as soon as the chloride concentration at the steel exceeds this threshold value. Literature reviews [5–9] revealed a huge scatter of values of C_{crit} , which is for a great part due to the wide variety of test methods used to determine C_{crit} , making it difficult to compare the results. Many of the literature results were obtained under laboratory conditions not representative for site conditions and are thus of little value for engineering purposes. In absence of practice-related data, the statistical distribution for C_{crit} used in service life modelling can generally be traced to one single laboratory study [5]. Thus, there is a lack of statistical information on practice-related C_{crit} (Fig. 1), e.g. C_{crit} values that are measured on samples from real structures and in realistic conditions. However, this is exactly what is needed in probabilistic service life modelling.

Moreover, drawing conclusions on the main factors affecting corrosion initiation is difficult, as the experimental conditions in the literature differ strongly. Thus, analysing the literature remains a challenge, because it is difficult to rule out the possible influence of different experimental methods used in different studies. A better documentation of C_{crit} and possible influencing factors (location of corrosion initiation with respect to the presence of defects at the steel-concrete interface, binder type, w/b-ratio, reinforcing steel properties, etc.) would significantly improve identifying the most dominant influencing factors for corrosion initiation.

2. A systematic data collection on chloride-induced corrosion

To this purpose a systematic collection of C_{crit} values together with a detailed documentation of the conditions at the steel-concrete interface and the location of corrosion initiation was recently established [10]. All C_{crit} reported in this data collection were measured with one experimental protocol. The data collection contains results from samples taken from by now 5 engineering structures. We consider it valuable that additional information about material properties of steel and concrete is also reported (steel microstructure, steel potential, depth of carbonation, concrete resistivity, binder type, etc.).

2.1. Critical chloride content C_{crit}

Due to the relatively high number of samples tested so far (in total

58 samples, on average ~ 10 per structure), the database allows the statistical treatment of C_{crit} results, which is needed for probabilistic service life modelling (see Section 3). A regularly updated version of the data collection can be freely accessed on the repository in Ref. [10]. For a more detailed description of the methodology used to collect the data it is referred to Ref. [11].

It is important to note that all the compiled C_{crit} values were determined with the same experimental method [12], thus ensuring comparability of the obtained results. This is important because the wide variety in experimental approaches used to measure C_{crit} in the literature is one of the main factors that to date hampers the systematic analysis of literature data [5–8,13]. Using one defined protocol allows overcoming this severe limitation of the current literature, and will allow to investigate the influence of different parameters, such as the w/b-ratio, the binder type, the rebar type, and the many more parameters that have been suggested to play a role in corrosion initiation.

It is also worth pointing out that the used protocol [12] to determine C_{crit} aimed at delivering as practice-related results as possible. This concerns the experimental test setup, electrochemical measurements, and the criterion to detect corrosion initiation. Most of the C_{crit} reported in this data compilation were determined on reinforced concrete cores retrieved from engineering structures, thus guaranteeing realistic material conditions for concrete and steel [10,11]. The samples were tested with the protocol described in [12], that essentially consists of exposing a defined area of the reinforced concrete specimen to chloride solution while monitoring the corrosion potential of the embedded steel.

The test protocol [12] used to determine C_{crit} was developed at ETH Zurich over the last years [14] and is to some extent based on the work of RILEM technical committee 235-CTC [15,16]. The aim during the development was to devise a test method that is capable of delivering realistic C_{crit} , viz. values that can be used in service life modelling.

2.2. Inspection of the steel-concrete interface

Upon corrosion initiation, the specimens were split to inspect the steel-concrete interface and to investigate the location of corrosion initiation with respect to any inhomogeneities at the steel-concrete interface. Thus in addition to C_{crit} , the data compilation contains a valuable systematic documentation of the location of corrosion initiation with respect to characteristics at the steel-concrete interface. Here, the effect of air voids and cracks at the steel-concrete interface and the influence of the non-carbonated cover depth on C_{crit} were investigated. The “non-carbonated cover depth” is the distance between the carbonation front (determined with the phenolphthalein spray test [17]) and the steel surface.

This systematic documentation of conditions at the steel-concrete interface and the point where corrosion occurred intends to contribute

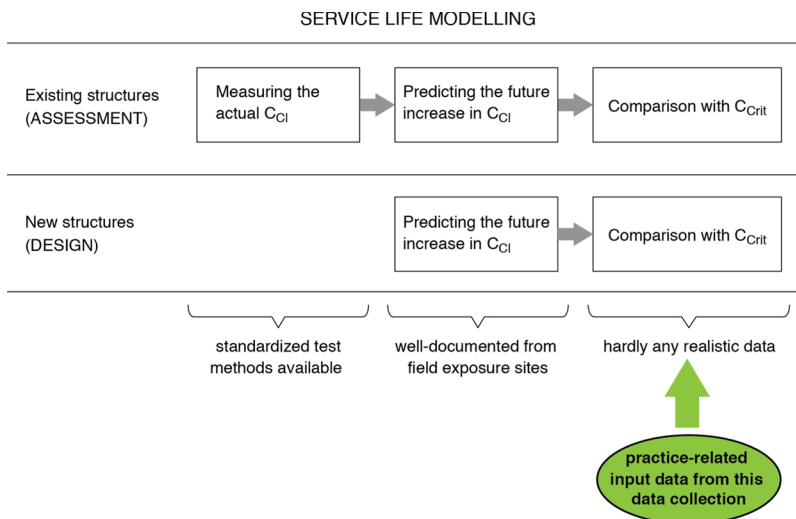


Fig. 1. Schematic illustration of service life modelling to predict the time of corrosion initiation for condition assessment and structural design. The time of corrosion initiation is reached when the actual chloride content C_{Cl} at level of reinforcement is equal or higher than the critical chloride content C_{crit} . The here presented database is providing realistic and significantly diverse input data for the model.

to a better understanding of the influence of local conditions at the steel-concrete interface on corrosion initiation, which is one of the major open questions in the field of corrosion of reinforced concrete structures [18] (compare Section 3.2).

2.3. Future application of the systematic data collection

It is worth mentioning that the data collection will be continuously updated in the future as new results become available (currently, more than 50 additional samples are being tested). In addition to our own experimental data, we are open to consider results from other research groups (to be added to the database) given that scholars followed the experimental protocol and guidelines described in [11,12]. Thanks to the open access nature of the data collection, the research community can immediately start exploring and analyzing the data. This may range from revealing unexpected, simple relationships between parameters to more complicated, multi-parameter analyses, e.g. by statistical methods or with help of machine learning, which are approaches that can typically not be easily applied to literature data due to the strong biases arising from different test methods used by different authors. In the following sections, we illustrate how this systematic data collection may contribute to improving service life modelling and understanding corrosion initiation in concrete.

3. Results and discussion

3.1. Utilizing the database of C_{crit} for service life modelling

3.1.1. Significance of input parameters on service life modelling

Predicting the time to corrosion initiation in concrete requires models to forecast over time the chloride content at a given depth, viz. the chloride content at the level of the embedded reinforcing steel (C_{Cl}). This numerical prediction of C_{Cl} is compared to C_{crit} . The time at which $C_{Cl} > C_{crit}$ is the time of corrosion initiation, which is often considered the service life. Service life modelling is on the one hand important for condition assessment of existing structures, where, based on the actual chloride content on the steel surface, the remaining service life can be estimated (Fig. 1, [19]). On the other hand, modelling the time to corrosion onset is used for the design of new structures, namely to ensure the durability of these within the design service life (Fig. 1).

The existing models describing chloride ingress into concrete often rely on empirical approaches using equations resembling diffusion laws [20–23]. Due to the variability inherent to both chloride ingress and C_{crit} , it is often preferred to use full probabilistic service life model approaches, that allow to forecast the probability for corrosion

initiation as a function of time. This requires that parameters for the chloride transport model – such as the chloride diffusion coefficient or the so-called ageing factor – as well as C_{crit} are considered as stochastic variables.

Selection of the stochastic input parameters of the model describing transport of chlorides ions through the concrete, based on variables such as cement type and concrete mix proportions, is possible based on numerous long-term field studies reporting data on chloride ingress into concrete [23–27] (Fig. 1). However, there is almost no stochastic data available for C_{crit} , particularly what concerns values representative for site-conditions [28]. The majority of the available laboratory experiments were conducted in solution or mortar rather than concrete, with smooth rather than ribbed steel and under often severely polarized conditions, which does not reflect realistic conditions. Thus it is common practice in service life modelling to use a unique distribution of C_{crit} (often the one reported in the fib model code for service life design [22]), regardless of the type of structure and materials used.

3.1.2. C_{crit} values representative for site-conditions

The novelty of this short communication are data on the critical chloride content from field structures. Fig. 2 shows examples of statistical distributions of C_{crit} that can be found in the current version of the data collection [10]. The distributions shown are from 5 different structures in Switzerland: 4 tunnels (for 2 of those tunnels, samples were taken from 2 different structural members (different walls) each) and 1 bridge (in which samples are distinguished between horizontal and vertical rebars). This yields 8 different tested series of samples. For each series the different samples tested were taken within a small area of the structure ($< 2 \text{ m}^2$). Thus the samples within one tested series were from presumably very similar material properties and exposure history. Despite this, Fig. 2 shows that the results can scatter over considerable ranges. Moreover, the value of C_{crit} determined can significantly differ from one structure to another and also from the distribution stipulated in the fib model code for service life design [22].

3.1.3. Methods for obtaining the statistical distribution of C_{crit} and modelling of service life

The experimental values for C_{crit} reported in the data collection (Fig. 2) can be described by statistical distribution functions, in order to be used as input data for probabilistic service life modelling. To illustrate this, the lognormal-distribution (cumulative distribution function CDF given in Eq. 1) was here used to describe the C_{crit} measured in 5 structures as shown in Fig. 2, with its parameters detailed in Table 1:

$$F_x(x) = \Phi\left(\frac{\ln x - \lambda}{\varepsilon}\right) \quad (1)$$

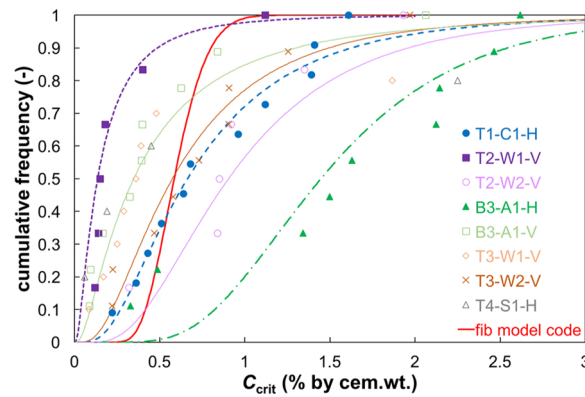


Fig. 2. Cumulative frequency of C_{crit} for eight different elements from five structures. T1-C1 indicates samples retrieved from a tunnel (T) ceiling (C); T2-W1 indicates samples taken from a tunnel wall (W); B3-A1 stands for the abutment wall (A) of a bridge (B). The last letters in each series indicates whether the rebars were horizontally (H) or vertically (V) oriented in the structure. The lines between the symbols are the fitted lognormal-distributions with the parameters given in Table 1. For comparison, also the distribution given in the fib model code for service life design [22] is plotted in red, which is a beta-distribution. The three lowest C_{crit} -values for T2-W1-V were measured in already carbonated concrete (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.).

where $\lambda = \mu_{\ln x}$ and $\varepsilon^2 = \sigma_{\ln x}^2$.

Modelling of the chloride ingress was conducted with the common empirical approach [29] (Eq. 2) to highlight the impact of using structure-dependent C_{crit} on service life modelling. A constant set of input data for the chloride ingress was used (given in Table 2), but different distributions of C_{crit} .

$$C(x, t) = C_s \times \left(1 - \operatorname{erf} \left(\frac{x}{2 \sqrt{D_{app} \times t}} \right) \right) \quad (2)$$

Here, $C(x, t)$ describes the chloride content at depth x and time t , C_s is the chloride surface concentration at the exposure side, and $D_{app}(t)$ the apparent chloride diffusion coefficient (shown in Eq. 3, age factor n and reference diffusion coefficient $D_{app,0}$ given in Table 2).

$$D_{app}(t) = D_{app,0} \times \left(\frac{t_0}{t} \right)^n \quad (3)$$

Depth x corresponds to the cover depth, chosen for all structures as 50 mm. The time to corrosion initiation is the time when $C(x, t)$ equals C_{crit} .

3.1.4. Predicted time to corrosion initiation for site-conditions

The probability of corrosion initiation was calculated for each field structure using its specific statistical distribution of the critical chloride content (Fig. 2, Table 1). The so calculated probability of corrosion initiation (Fig. 3) varies strongly for each different field structure, even if the same chloride ingress parameters are chosen. Depending on the considered structure, the predicted time to corrosion initiation can be significantly shorter or longer than the prediction assuming C_{crit} as stipulated in the fib model code for service life design. Moreover, the example calculations shown in Fig. 3 indicate that C_{crit} is likely to have a more dominant influence on the predicted time to corrosion initiation than the chloride transport properties. This highlights the decisive

Table 1

Parameters of fitted lognormal-distribution for the three C_{crit} datasets shown in Fig. 2. The parameters for the beta-distribution of the fib model code is also given. A and B are the lower and upper border of the beta distribution, respectively.

Parameter	T1-C1-H	T2-W1-V	T2-W2-V	B3-A1-H	B3-A1-V	T3-W2-V	fib model code [22]
λ	-0.41	-2.0	-0.1	0.37	-1.06	-0.55	-
ε	0.70	1.0	0.6	0.44	1.03	0.75	-
α	-	-	-	-	-	-	5.3
β	-	-	-	-	-	-	18.6
A	-	-	-	-	-	-	0.2
B	-	-	-	-	-	-	2

Table 2

Parameters for the chloride ingress model example.

Parameter for all structures		
cover depth (mm)	x	50
diffusion coefficient at reference time t_0 (m^2/s)	$D_{app,0}$	2×10^{-12}
age coefficient (-)	n	0.3
surface chloride concentration (M-% by cem.wt.)	C_s	2

influence of the C_{crit} on the results of service life modelling, in agreement with other studies [30].

3.1.5. Outlook for future service life modelling

The continuously growing data compilation for C_{crit} , including corresponding material properties (concrete type, steel type, etc.) and structural details (age, exposure, etc.), will considerably broaden the array of available input data for service life modelling. We are confident that this will permit selecting more appropriate, viz. more realistic, model input values for C_{crit} than the current approach of using tabulated unique values. Thus the predictive power of service life models will improve, as the input values are more practice-related.

3.2. Towards understanding the influence of the steel-concrete interface on corrosion initiation

Besides C_{crit} , the data compilation contains a systematic documentation of the location of corrosion initiation with respect to characteristics at the steel-concrete interface to study the influence of characteristics at the steel-concrete interface on the C_{crit} . To highlight this benefit, the example of the influence of air voids and the pH of the pore solution on the C_{crit} is briefly discussed.

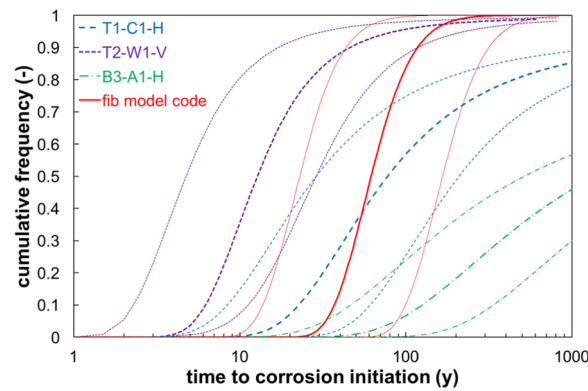


Fig. 3. The cumulative frequency of corrosion initiation vs. time for three different cases (C_{crit}) shown in Fig. 2. The chloride ingress was modelled with the error function and parameters given in Table 2. The area between the thinner lines shows the influence of the chloride ingress parameters (diffusion coefficient from Table 2 was varied by the factor 2).

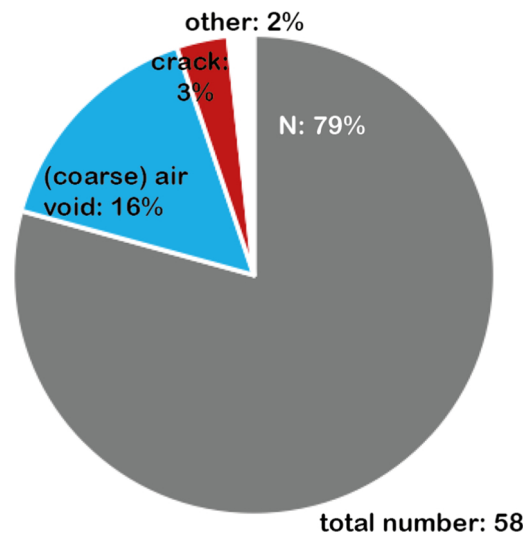


Fig. 4. The frequency of occurrence of corrosion initiation at different local conditions at the concrete interface. In 79% of the in total 58 evaluated cases, no irregularities were visually apparent at the location of corrosion at steel-concrete interface (N, dark grey). The blue colour indicates the portion of cases where corrosion initiated at a coarse air void. Only in a minority of cases did corrosion occur at cracks (red; here, cracks mean small cracks at the steel-concrete interface that are not expanding to the exposure side of the concrete sample) (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.).

3.2.1. Influencing parameters at the steel-concrete interface on C_{crit} – air voids

The data in Fig. 4 shows that in most cases (approx. 80%), no irregularities were visually apparent at the location of corrosion initiation at the steel-concrete interface. In approx. 20% of the cases, the corrosion spot was located at irregularities, including coarse air voids, cracks, or tie wires. Fig. 5 depicts the influence of these defects on the distribution of C_{crit} . Interestingly, statistically significantly higher chloride contents (level of significance: 20%) were needed to initiate corrosion at coarse air voids than at locations with no visually apparent irregularities at the steel-concrete interface. These results may appear to be in contradiction with other studies, where interfacial air voids are typically found as preferential sites for corrosion initiation [31,32]. This raises an important scientific question that still needs to be answered by further research. Current discussions in RILEM technical committee 262-SCI [18] have led to the hypothesis that this apparent disagreement can be explained by the moisture state of macroscopic voids. The air voids in concrete, exposed in splash water situations such as exposure regime XD3 [33] are generally not water-saturated [34]. It is hypothesized that in largely air-filled macroscopic voids, corrosion initiation is hardly possible due to the absence of an electrolyte. In this work, the concrete samples were exposed in solution, with merely short

intervals of drying in the lab twice a year. The relative humidity (RH) in the concrete at the depth of the rebars was always between 95–98% (measured with RH sensors in companion samples). This confirms that air voids were here unsaturated [34,35], which may explain the observation that coarse air voids were in the majority of cases not favourable locations for corrosion initiation (Figs. 4, 5).

3.2.2. Influencing parameters at the steel-concrete interface on C_{crit} – pH at the steel surface

Fig. 6 shows a strong correlation between the non-carbonated cover depth and the C_{crit} . The C_{crit} is increasing with increasing non-carbonated cover depth. This can be explained by the pH at the steel surface, which depends on how close the carbonation front has approached to the steel. The non-carbonated cover depth is an indirect indication of the pH of the pore solution at the level of reinforcement. This is because the pH is not decreasing immediately from $pH > 13$ to approx. 9 at the carbonation front, but shows a more gradual decrease [36]. Thus, also ahead of the carbonation front, the pH is lower than the initial highly alkaline value. The significance of the pH on chloride-induced corrosion initiation is discussed in Ref. [37]. The change of the pH influences the ratio between free and bound chlorides [38,39], leading therefore to a higher concentration in free chlorides. Certainly, this increases the risk

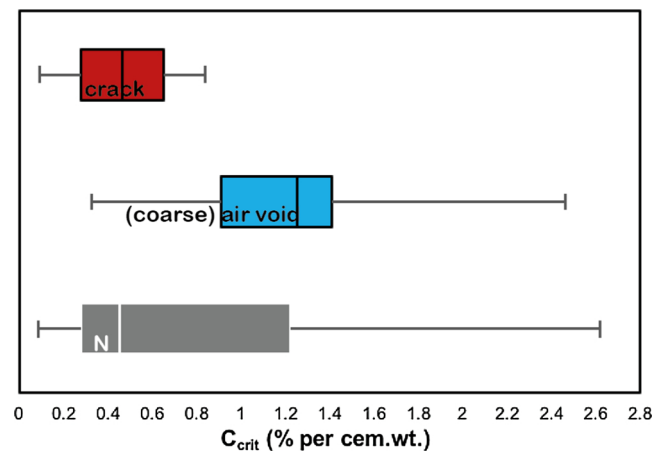


Fig. 5. Box plots of C_{crit} for the 58 evaluated cases of corrosion initiation related to the steel-concrete interface characteristics (see Fig. 4 for terminology). The boxes indicate the 25%, 50%, and 75%-quantiles, and the whiskers are extreme values. For cracks (red) only two values were documented so far (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.).

for corrosion initiation for the same total chloride content.

3.2.3. Outlook for further understanding of the mechanism of corrosion initiation

Analysing the factors that influence corrosion initiation in concrete is complex, due to the almost countless parameters that may potentially influence corrosion initiation. Thus, a complete analysis is beyond the scope of this short communication. We are convinced that by making the comprehensive and systematically collected data freely accessible to the research community, scholars may use different approaches and techniques for refined analyses. This will promote further discussion and enables corrosion scientists working on chloride-induced corrosion of reinforcing steel in concrete to further understand the mechanism of corrosion initiation in concrete.

4. Concluding remarks and perspectives

In line with current open data initiative this short communication presents for the first time a systematic data collection of critical chloride content data from engineering structures obtained with the same test protocol [10]. It documents and highlights the importance of a site-specific statistical distribution of the critical chloride content C_{crit} for more reliable probabilistic service life modeling and intends to stimulate discussion on this important topic among scholars working in this field. The main conclusions are:

1) The statistical distribution of C_{crit} can significantly differ from one

structure to another and even between members within one structure. Since a wide range of possible influencing factors – such as type of structure, concrete and steel properties and characteristics at the steel-concrete interface – are also documented in the data collection, engineers performing service life modelling now have a basis to select a more adequate statistical distribution as input data for the existing models than by assuming a constant distribution, such as the one given in the fib model code for service life design [22].

- 2) Systematically reporting C_{crit} together with corresponding potential influencing parameters will enhance the understanding and promote future discussion of the without doubt important questions related to the mechanism of chloride-induced corrosion initiation in concrete. The proposed data collection allows studying the influence of local conditions at the steel-concrete interface on samples from real structures, which are known to show considerably different characteristics at the steel-concrete interface than samples produced in the laboratory [35].
- 3) Our results indicate that the role of air voids in promoting corrosion initiation is more complex than previously expected and needs additional considerations such as the moisture state of the concrete. Additionally, pH changes ahead of the carbonation front were found to significantly influence the susceptibility to corrosion. We recommend further studies to address these important questions.

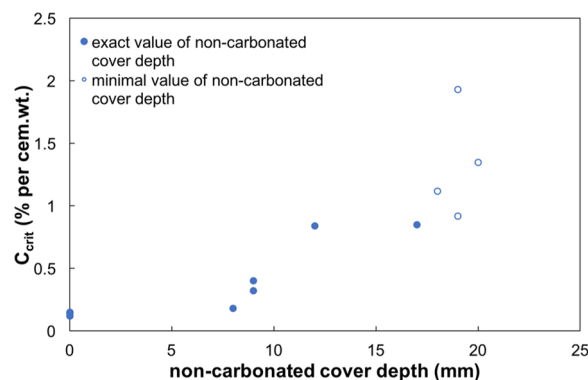


Fig. 6. Comparison of the non-carbonated cover depth with C_{crit} of one selected structure. The filled circles represent the exact value of non-carbonated cover depth, measured directly at the split sample with Phenolphthalein. The empty circles indicate minimal values of the non-carbonated cover depth. In these cases, the analysed part of the cover concrete did not show any carbonation and the actual non-carbonated cover depth remains unknown (but certainly higher than the given value).

Data availability

Raw and processed data are described in detail in a “Data in Brief” paper [11]. The raw data required to reproduce these findings are available to download from [10].

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